Final Technical Progress Report

# Palladium/Copper Alloy Composite Membranes for High Temperature Hydrogen Separation from Coal-Derived Gas Streams

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Submitted by

J. Douglas Way Chemical Engineering and Petroleum Refining Department Colorado School of Mines Golden, CO 80401-1887

> Office: (303) 273-3519 Telefax: (303) 273-3730 EMAIL: dway@mines.edu

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Administrative Contact:

Mary Mittag-Miller (303) 273-3411

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#### ABSTRACT

For hydrogen from coal gasification to be used economically, processing approaches that produce a high purity gas must be developed. Palladium and its alloys, nickel, platinum and the metals in Groups 3 to 5 of the Periodic Table are all permeable to hydrogen. Hydrogen permeable metal membranes made of palladium and its alloys are the most widely studied due to their high hydrogen permeability, chemical compatibility with many hydrocarbon containing gas streams, and infinite hydrogen selectivity. Our Pd composite membranes have demonstrated stable operation at 450 °C for over 70 days. Coal derived synthesis gas will contain up to 15000 ppm H<sub>2</sub>S as well as CO, CO<sub>2</sub>, N<sub>2</sub> and other gases. Highly selectivity membranes are necessary to reduce the H<sub>2</sub>S concentration to acceptable levels for solid oxide and other fuel cell systems. Pure Pd-membranes are poisoned by sulfur, and suffer from mechanical problems caused by thermal cycling and hydrogen embrittlement. Recent advances have shown that Pd-Cu composite membranes are not susceptible to the mechanical, embrittlement, and poisoning problems that have prevented widespread industrial use of Pd for high temperature  $H_2$  separation. These membranes consist of a thin ( $\leq 5 \,\mu$ m) film of metal deposited on the inner surface of a porous metal or ceramic tube. With support from this DOE Grant, we have fabricated thin, high flux Pd-Cu alloy composite membranes using a sequential electroless plating approach. Thin,  $Pd_{60}Cu_{40}$ films exhibit a hydrogen flux more than ten times larger than commercial polymer membranes for H<sub>2</sub> separation, resist poisoning by H<sub>2</sub>S and other sulfur compounds typical of coal gas, and exceed the DOE Fossil Energy target hydrogen flux of 80 ml/cm<sup>2</sup>•min =0.6 mol/m<sup>2</sup>•s for a feed pressure of 40 psig. Similar Pd-membranes have been operated at temperatures as high as 750 C. We have developed practical electroless plating procedures for fabrication of thin Pd-Cu composite membranes at any scale.

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## **INTRODUCTION AND OBJECTIVES**

Recent advances have shown that Pd-Cu composite membranes are not susceptible to the mechanical, embrittlement, and poisoning problems that have prevented widespread industrial use of Pd for high temperature H<sub>2</sub> separation. These membranes consist of a thin (~1  $\mu$ m) film of metal deposited on the inner surface of a porous metal or ceramic tube. Based on preliminary results, thin Pd<sub>60</sub>Cu<sub>40</sub> films are expected to exhibit hydrogen flux up to ten times larger than commercial polymer membranes for H<sub>2</sub> separation, and resist poisoning by H<sub>2</sub>S and other sulfur compounds typical of coal gas. Similar Pd-membranes have been operated at temperatures as high as 750 C. The overall objective of the proposed project is to demonstrate the feasibility of using sequential electroless plating to fabricate Pd<sub>60</sub>Cu<sub>40</sub> alloy membranes on porous supports for H<sub>2</sub> separation. These following advantages of these membranes for processing of coal-derived gas were demonstrated:

High  $H_2$  flux, Sulfur tolerant, even at very high total sulfur levels (1000 ppm), Operation at high temperatures, Resistance to embrittlement and degradation by thermal cycling.

The objectives of this research project were to provide a fundamental understanding of:

Factors important in membrane fabrication. Optimization of membrane structure and composition. Effect of temperature, pressure, and gas composition on  $H_2$  flux and membrane selectivity. How this membrane technology can be integrated in coal gasification-fuel cell systems.

## **EXECUTIVE SUMMARY**

For hydrogen from coal gasification to be used economically, processing approaches that produce a high purity gas must be developed. Palladium and its alloys, nickel, platinum and the metals in Groups 3 to 5 of the Periodic Table are all permeable to hydrogen. Hydrogen permeable metal membranes made of palladium and its alloys are the most widely studied due to their high hydrogen permeability, chemical compatibility with many hydrocarbon containing gas streams, and infinite hydrogen selectivity. Our Pd composite membranes have demonstrated stable operation at 450 °C for over 70 days. Coal derived synthesis gas will contain up to 15000 ppm H<sub>2</sub>S as well as CO, CO<sub>2</sub>, N<sub>2</sub> and other gases. Highly selectivity membranes are necessary to reduce the H<sub>2</sub>S concentration to acceptable levels for solid oxide and other fuel cell systems. Pure Pd-membranes suffer from mechanical problems caused by thermal cycling and hydrogen embrittlement. Recent advances have shown that Pd-Cu composite membranes are not susceptible to the mechanical, embrittlement, and poisoning problems that have prevented widespread industrial use of Pd for high temperature H<sub>2</sub> separation. These membranes consist of a thin ( $\leq 5 \mu m$ ) film of metal deposited on the inner surface of a porous metal or ceramic tube.

With support from this DOE Grant, we have fabricated thin, high flux Pd-Cu alloy composite membranes using a sequential electroless plating approach. Thin  $Pd_{60}Cu_{40}$  films that exhibit a hydrogen flux more than ten times larger than commercial polymer membranes for H<sub>2</sub> separation, resist poisoning by H<sub>2</sub>S and other sulfur compounds typical of coal gas, and **exceed** the DOE Fossil Energy target hydrogen flux of 80 ml/cm<sup>2</sup>•min =0.6 mol/m<sup>2</sup>•s for a feed pressure of 40 psig. Similar Pd-membranes have been operated at temperatures as high as 750 C. We have developed practical electroless plating procedures for fabrication of thin Pd-Cu composite membranes at any scale.

We have shown that a 40 weight % Cu binary alloy with Pd has the highest permeability, consistent with the literature on Pd alloy foils. Preliminary results have shown that although exposure to H<sub>2</sub>S inhibits (reduces) the H<sub>2</sub> flux, this effect is reversible. Permeation measurements with mixtures representative of synthesis gas show that the H<sub>2</sub> flux is not inhibited by the presence of CO. We have also made a membrane that exhibits very high ideal H<sub>2</sub>/N<sub>2</sub> selectivity,  $\geq$  7000.

#### EXPERIMENTAL

Composite alloy membranes were fabricated by sequential electroless plating of palladium and copper onto symmetric 0.2  $\mu$ m cut-off  $\alpha$ -alumina tubes (CoorsTek GTC-998) and asymmetric 0.05 and 0.02  $\mu$ m cut-off zirconia coated  $\alpha$ -alumina tubes (US Filter T1-70). A seeding procedure has to be used prior to the palladium plating in order to insure adhesion between the metallic film and the ceramic surface [1]. That step involves impregnation of the ceramic support using an organic Pd salt solution, followed by calcination and reduction in flowing hydrogen.

Pd and Cu electroless plating baths from the literature were then used in combination with osmotic pressure gradients to deposit films ranging from 1 to 25  $\mu$ m in thickness [2]. The osmotic pressure, generated by circulating concentrated sucrose solutions on the outside of the tubes, insured reduced porosity and promoted surface homogeneity and densification of the plated Pd film [3]. After sequential plating of Pd and Cu films, the membranes were annealed during hydrogen permeation tests at temperatures greater than 350 °C to produce homogeneous alloy membranes.

## **RESULTS AND DISCUSSION**

Membranes with metal layer as thin as 1 micron were made routinely. Figure 1 shows an SEM image of a Pd-Cu composite membrane with a metal film roughly 1 µm thick. These membranes have very high fluxes as shown in Figure 2. However, it was found that the required metal film thickness to produce a leak-free membrane was a function of the plating time as well as being influenced by the support choice [4, 5]. In the absence of structural defects in the support, the nominal pore size and smoothness of the top layer are the most important factors determining the minimal thickness to produce a leak free membranes to be made, but the adhesion, or penetration of the metal film in to the support, is decreased. Consequently, a balance must be found to minimize surface roughness while maintaining adequate adhesion of the Pd alloy film to the support.

The fluxes for these very thin Pd-Cu composite membranes **exceed** the DOE Fossil Energy program goal for hydrogen flux of 80 ml/cm<sup>2</sup>•min =0.6 mol/m<sup>2</sup>•s for a feed pressure of 40 psig [6]. As can be seen in Figure 2 below, the flux for a pure hydrogen feed pressure of 40 psig = 52 psia is ~ 0.8 mol/m<sup>2</sup>•s = 107 ml/cm<sup>2</sup>•min at the moderate temperature of 350 °C.

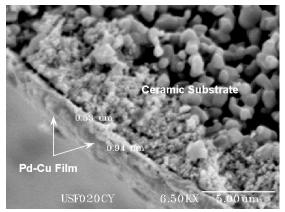


Figure 1. SEM picture of Pd-Cu composite membrane #20, 90 mass % Pd, on a 20 nm cut-off asymmetric ceramic filter. Scale bar is 5  $\mu$ m.

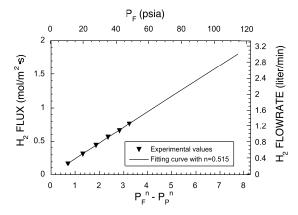


Figure 2. Influence of driving force on the pure gas  $H_2$  flux for a 1.5  $\mu$ m thick Pd-Cu membrane (#25b) having 60 mass % Pd at 350 °C.  $P_F$  is feed pressure.

## **Effect of Pd Alloy Composition**

The composition of the Pd-Cu alloy film has a strong influence on the  $H_2$  permeability. Figure 3 shows that the highest  $H_2$  permeability is observed for our composite membranes at a alloy composition of 60 mass %. This is consistent with the patent literature for Pd-Cu foil membranes [7]. Furthermore, the Pd inventory of this alloy and resulting cost is significantly less than the Pd-Ag alloy, for example, which has a maximum  $H_2$  permeability at 77 mass % Pd.

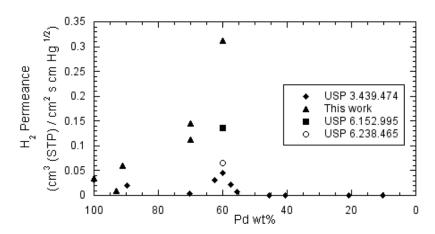


Figure 3. The influence of alloy composition on the pure  $H_2$  permeance for Pd-Cu composite membranes at 350 °C. The Figure compares our composite membranes to the performance of Pd-Cu foil membranes from the patent literature [7-10]

#### Effect of H<sub>2</sub>S Exposure

Figure 4 below shows the influence of  $H_2S$  exposure on the pure gas  $H_2$  flux for a Pd-Cu membrane. After annealing in He for 5 days, membrane USF-AK-20-9 was exposed to 100 ppm  $H_2S$  in an 8 hr test. Before the  $H_2S$  exposure, the membrane had a steady state flux of 0.137 mol/m<sup>2</sup> s and  $H_2/N_2$  ideal selectivity of 23 at 450°C and 40 psig feed pressure. The addition of 100 ppm  $H_2S$  to a mixture of 50/50  $H_2/N_2$  balance for 8 hours decreased the  $H_2$  flux from 0.137 to 0.081 mol/s/m<sub>2</sub> and ideal selectivity from 23 to 11. Although after the first one hour of  $H_2S$  introduction, the flux through the membrane stabilizes, this indicates a ~40% reduction in the

hydrogen flux due to sulfur inhibition. However, it is anticipated that in the presence of steam, the inhibition of the H<sub>2</sub> flux due to H<sub>2</sub>S exposure will be less severe. Following the H<sub>2</sub>S experiment, a 20% H<sub>2</sub>O/H<sub>2</sub> mixture was used to sweep H<sub>2</sub>S from the membrane for 2 hour. The 20% H<sub>2</sub>O/H<sub>2</sub> sweep increased the H<sub>2</sub> flux to 0.093 mol/m<sup>2</sup> s and the ideal selectivity to 19. Figure 1 presents these results in a graphic form. In addition, we also observed an increase in the ideal selectivity to 28 by purging the membrane with He overnight after the steam exposure. Air oxidations (described in detail in the Proposed Research section) will also reverse the effects of H<sub>2</sub>S exposure.

These preliminary data suggest that our Pd-Cu membranes, produced by sequential electroless plating, are **not poisoned** by exposure to  $H_2S$ . Further research is needed to examine the effects of Cu composition in the Pd-Cu alloy, the effect of  $H_2S$  concentration in the feed gas, and regeneration procedures such as air or steam oxidations.

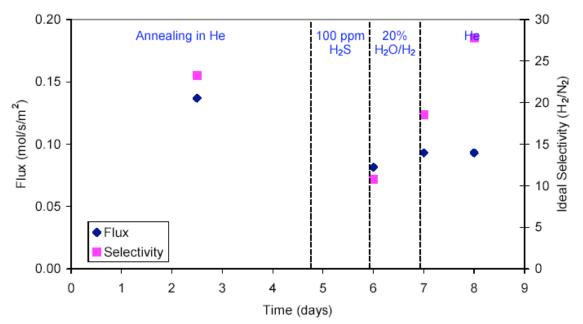


Figure 4. Effect of  $H_2S$  exposure on  $H_2$  single gas flux and  $H_2/N_2$  ideal selectivity at 40 psig and 450 C.

## **Effect of Feed Gas Mixture**

Recently, we tested our Pd-Cu membrane performance in the presence of a model water gas shift (WGS) mixture gases having the following mixture concentration: 42% H<sub>2</sub>, 19% CO<sub>2</sub>, 4% CO and 35% H<sub>2</sub>O, on volume basis. As shown in Figure 5, keeping the hydrogen partial pressure driving force constant, the hydrogen flux across the membrane for a 50 psig WGS feed mixture at 450 °C, is approximately the same as that observed for a 42% H<sub>2</sub>/N<sub>2</sub> binary (shown in the first 45 minutes).

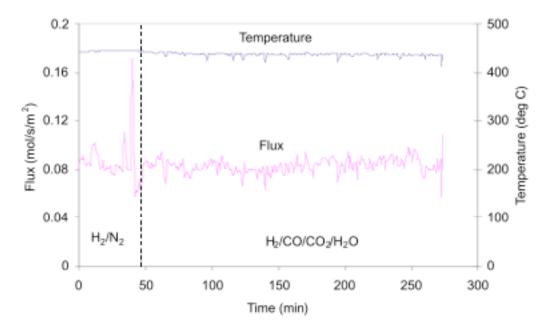


Figure 5. Flux versus time after introduction of WGS synthesis gas to membrane USF-AK-50-6 at 50 psi and 450 C.

#### Fabrication of High Selectivity Pd-Cu Membrane

Figure 6 shows the complete data set for membrane #28, a PdCu film (8% Cu, balance Pd) on a ceramic filter support. Until the "air purge" (controlled oxidation) treatment at 69 days, the  $H_2/N_2$  selectivity of the membrane was at least 7000, there was no measurable permeation of nitrogen. After reaching a steady-state flux at a temperature of 450 °C, with a hydrogen feed pressure of 50 psig, the temperature was reduced to 400 °C and then 350 °C to compare with earlier data. As expected the  $H_2$  fluxes at these temperatures were higher than measured earlier in the test due to annealing of the Pd and Cu phases. The highest  $H_2$  flux of 0.35 mole/m<sup>2</sup>•s = 0.78 cm<sup>3</sup>(STP)/cm<sup>2</sup>•s was observed at 450 °C after 50 days of testing.

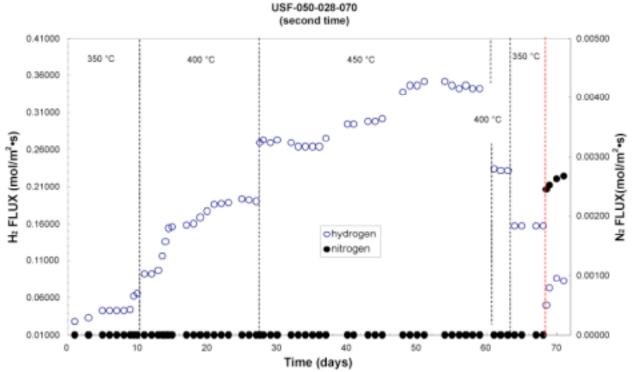


Figure 6. Complete data set for membrane #28 for a feed pressure of 50 psig. Atmospheric pressure in Golden, CO is 12 psia. The black dotted lines denote change in temperature and the red dotted line represents an "air purge" treatment. Until the air purge treatment (red dotted line), the selectivity of the membrane was at least 7000, there was no measurable permeation of nitrogen. Immediately after the air purge the membrane developed a large inert gas leak and the selectivity dropped to approximately 30.

## Effect of Membrane Surface Structure.

We have observed an unexpected flux increase for our membranes following exposure to air at high temperature and subsequent reduction of the Pd oxide surface. For example, as shown in Figure 7, "air purge" treatments at 6, 8, and 13 days caused a sharp increase in the H<sub>2</sub> flux. At 8 days, the H<sub>2</sub> flux sharply increased to 0.7 mol/m<sup>2</sup>•s and then decreased to a new steady-state value of 0.5 mol/m<sup>2</sup>•s, 47% higher than the flux before the air purge treatment. The air purge was repeated for a third time at 13 days.

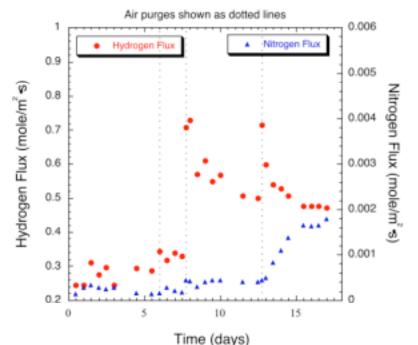


Figure 7. Flux data for Membrane #20, 90 wt. % Pd. Film thickness  $\approx 1 \mu m$ , temperature = 350 C. The pressure differential for the permeation tests was 50 psig. Note the flux increases due to the "air purge" treatments, shown as a dotted lines.

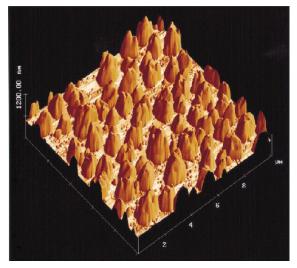
The  $H_2$  flux increase is very similar, but this time the  $N_2$  flux increased dramatically. Further examination of the data show that the  $N_2$  did increase after each air purge treatment.

Figure 8 is a cartoon that presents our hypothesis for the mechanism of the air purge treatment. If the film rearranged due to the air purge, then leak paths could be formed between the conical structures formed by the larger volume of the oxide. The increased  $H_2$  flux could be due to increased surface area and/or unknown chemical effects. There is evidence in the nanotechnology literature to support this hypothesis. Figure 9 shows conical palladium oxide structures formed by oxidizing a Pd film on a single crystal oxide support at 900 °C [11]. The height of these structures is approximately 1  $\mu$ m, close to the thickness of our Pd-Cu films. We



Figure 8. Cartoon showing our hypothesis for mechanism of air purge on thin Pd alloy membranes.

have shown (Figure 10) that a Pd film on a Si wafer formed by electroless deposition will show some similar conical structures upon oxidation but the temperatures are much higher than those used during air purge treatments. We intend to use AFM and XPS to further investigate this interesting and useful phenomenon in the future. AFM software can calculate the surface area of a sample, allowing us to plot flux versus surface area to see if the flux increases we observe are simply due to an increase in the number of sites for  $H_2$  dissociation.



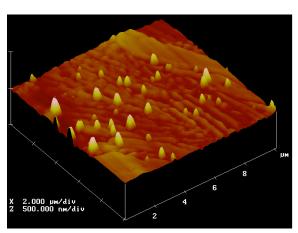


Figure 9. AFM image of Palladium oxide nano-structures formed by oxidation of a Pd film at 900 °C (Aggarwal et al., 2000).

Figure 10. AFM image of the surface of an electrolessly deposited Pd film on a Si wafer oxidized at 900 °C.

## TECHNOLOGY TRANSFER ACTIVITY

The following journal articles, presentations, and patents were made at technical meetings related to this project:

#### **Journal Articles**

Roa, F., Block, M. J., and J. D. Way, "The Influence of Alloy Composition on the H<sub>2</sub> Flux of Pd-Cu Composite Membranes," Desalination, 147, 411-416(2002).

Paglieri, S. N. and J. D. Way, "Innovations in Pd Membrane Research," Separation and Purification Methods, 31(1), 1-169(2002).

Roa, F. Way, J. D., McCormick, R. L. and S. N. Paglieri, "Preparation and Characterization of Pd-Cu Composite Membranes for Hydrogen Separation," The Chemical Engineering Journal, in press, 6/2002.

Road, F. and J. D. Way, "The Pressure Dependence of the Hydrogen Flux of Palladium-Copper Composite Membranes, in preparation, 2003.

## Presentations

ICOM 99,6/99, Toronto, Canada, Pd/Cu Composite Membranes for H<sub>2</sub> Separations (talk given by Steve Paglieri)

DOE-UCR Contractor's Meeting, Pittsburgh, PA 6/00, Pd-Cu Alloy Composite Membranes for Hydrogen Separations

Los Alamos National Laboratory, Los Alamos, NM, 12/00, New Materials for Gas and Liquid Separations

ACS National Meeting, San Diego, CA, 4/01, Micron Scale Pd-Cu Alloy Composite Membranes for Hydrogen Separations

DOE-UCR Contractor's Meeting, Pittsburgh, PA 6/01, Pd-Cu Alloy Composite Membranes for Hydrogen Separations

The North American Catalysis Society annual meeting, Toronto, Canada, 6/01, Scale Pd-Cu Alloy Composite Membranes for Hydrogen Separations and Membrane Reactors

Idaho National Engineering and Environmental Laboratory, Idaho Falls, ID, 7/01, New Materials for Gas and Liquid Separations

ICOM 02, Toulouse, France, 6/02, The Influence of Alloy Composition on the H<sub>2</sub> Flux of Pd-Cu Composite Membranes (talk given by Fernando Roa)

#### Patents

Way, J. D., Roa, F. and S. N. Paglieri, "Process for Preparing Palladium Alloy Composite Membranes for Use in Hydrogen Separation, Palladium Alloy Composite Membranes and Products Incorporating or Made From the Membranes," U. S. Patent Application, April 3, 2002.

## CONCLUSIONS

Our previous results have shown that we can fabricate thin, high flux Pd-Cu alloy composite membranes using a sequential electroless plating approach. We have shown that a 40 weight % Cu binary alloy with Pd has the highest permeability, consistent with the literature on Pd alloy foils. As shown in Figure 2, a 1.5  $\mu$ m thick Pd<sub>60</sub>Cu<sub>40</sub> membrane meets the DOE hydrogen flux target. Preliminary results have shown that although exposure to H<sub>2</sub>S inhibits (reduces) the H<sub>2</sub> flux, this effect is reversible. Permeation measurements with mixtures representative of synthesis gas show that the H<sub>2</sub> flux is not inhibited by the presence of CO. We have also made a membrane that exhibits very high ideal H<sub>2</sub>/N<sub>2</sub> selectivity,  $\geq$  7000.

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- 2. Roa, F., M.J. Block, and J.D. Way, *The influence of alloy composition on the H*<sub>2</sub>*flux of composite Pd–Cu membranes.* Desalination, 2002. 147: p. 411-416.
- 3. Yeung, K.L., J.M. Sebastian, and A. Varma, *Novel preparation of Pd/Vycor composite membranes*. Catal. Today, 1995. 25: p. 231-236.
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